

Melting of “Porous” Vortex Matter

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Bitter decoration and magneto-optical studies reveal that in heavy-ion irradiated superconductors, a “porous” vortex matter is formed when vortices outnumber columnar defects. In this state ordered vortex crystallites are embedded in the “pores” of a rigid matrix of vortices pinned on columnar defects. The crystallites melt through a first-order transition while the matrix remains solid. The melting temperature increases with density of columnar defects and eventually turns into a continuous transition. At high temperatures a sharp kink in the melting line is found, signaling an abrupt change from crystallite melting to melting of the rigid matrix.

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Melting of heterogeneous systems, and, in particular, of nanocrystals embedded in porous rigid matrices, is a complex process with many uncontrolled parameters. Metal and semiconductor nanocrystals with free surfaces usually show a decrease in their melting temperature T_m with decreasing size [1], whereas nanocrystals encapsulated in a porous matrix often display an increase in T_m [2]. This melting process is known to depend on the size, dimensionality, material properties of the nanocrystals and the matrix, as well as the interface energies between the materials [1,2]. In this work we investigate an analogous, but more controllable composite system, which is a “porous” vortex matter consisting of vortex nanocrystals encapsulated in a matrix of strongly pinned vortices. As shown below, this system is present in the commonly heavy-ion irradiated superconductors when the vortices outnumber the columnar defects (CDs). The rigid matrix is created by vortices localized on the network of random CDs, while the softer nanocrystals are formed within the pores of this matrix by the interstitial vortices. The size of the nanocrystals can be readily varied from several hundred down to a few vortices by changing the applied field or the density of CDs. We find that this composite vortex matter reveals a number of intriguing mechanisms: Similarly to the metallic nanocrystals in a matrix, we observe for the first time a pronounced upward shift in the vortex melting temperature T_m , while *preserving* the first-order nature of the transition (FOT). With increasing density of CDs, the size of the pores decreases, resulting in a larger shift in T_m . We also find a critical point at which the FOT changes into a continuous melting. Moreover, the crystallites can melt while the matrix remains rigid, giving rise to an abrupt breakdown in the upward shift of T_m and a sharp kink in the FOT line.

The reported findings were obtained using Bitter decoration and differential magneto-optical (MO) [3] tech-

niques. High quality $\text{Bi}_2\text{Sr}_2\text{CaCu}_2\text{O}_8$ (BSCCO) crystals ($T_c \approx 89$ K) were irradiated by 1 GeV Pb ions through various patterned masks at GANIL, with doses corresponding to matching fields of $B_\phi = 5, 10, 20,$ and 50 G. Figure 1(a) shows schematically one of these masks which results in the formation of CDs only within the circular apertures of about $90 \mu\text{m}$ diam. This patterning allows very sensitive comparison of the vortex structure and the melting processes in adjacent irradiated and pristine regions, which is not possible by other methods.

In the absence of CDs the vortices form quasicrystalline lattice with no topological defects (Bragg glass) [4], as seen in the pristine part of the magnetic decoration image in Figs. 1(b) and 1(c). What happens to this phase when sparse CDs are added? The irradiated region in Fig. 1(b)

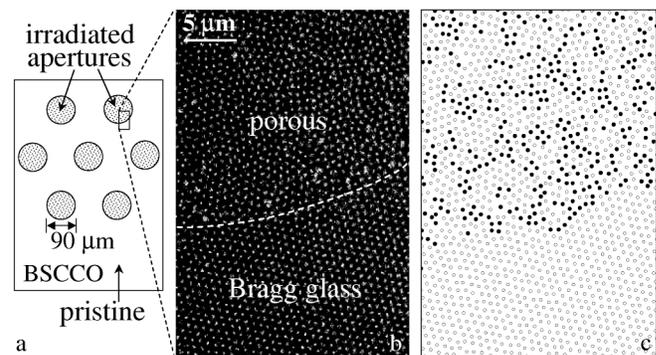


FIG. 1. (a) Schematic of BSCCO crystal irradiated through a mask with an array of circular apertures. (b) Bitter decoration image ($B = 40$ G, $T = 4.2$ K) showing pristine region at the bottom (Bragg glass) and a section of the irradiated region ($B_\phi = 10$ G) on top where porous vortex matter is formed. (c) Corresponding locations of sixfold (open circles) and of five- and sevenfold coordinated vortices (solid circles) obtained by Delaunay triangulation.

shows that it is no longer Bragg glass since it has significant amount of topological defects [solid circles in Fig. 1(c)] and no orientational long-range order [5]. It is also not an amorphous or glass phase in the usual sense, nor is it a simple polycrystal as discussed below. In the presence of CDs the Bose glass (BG) theory [6–8] is usually applied, which describes the vortex matter in terms of anisotropic *homogeneously* pinned medium. Such a description is adequate for the common situation of high irradiation doses ($B_\phi > B$), in which all the vortices reside on CDs and the vortex pinning energies are comparable. In contrast, we investigate here mainly the opposite extreme of low doses in which the vortices greatly outnumber the CDs, i.e., $B \gg B_\phi$. In this case the system is inherently *heterogeneous*, consisting of two vortex populations with *well separated characteristic energies*: The vortices residing on CDs are strongly pinned and form a rigid network or matrix, whereas the interstitial vortices are localized by significantly weaker elastic interactions and form relatively soft crystallites within the pores of the matrix. In the following we refer to this state [upper parts of Figs. 1(b) and 1(c)] as porous vortex matter in order to emphasize the important consequences of the heterogeneous structure.

Figure 2 shows several frames from a “movie” [9] of the melting process at indicated temperatures T at two fields, 30 and 60 G. Each frame is obtained by taking the difference between the MO images at $T + 0.15$ K and $T - 0.15$ K and averaging a large number of such differential images [3]. The bright features show the regions in the sample that undergo a FOT within the temperature interval of 0.3 K, and the intensity of this bright paramagnetic signal is the equilibrium magnetization step ΔB at the transition [3,10]. Figure 2(a) shows the nucleation of the liquid phase in the form of bright inclined strips in the central pristine region which arise from intrinsic

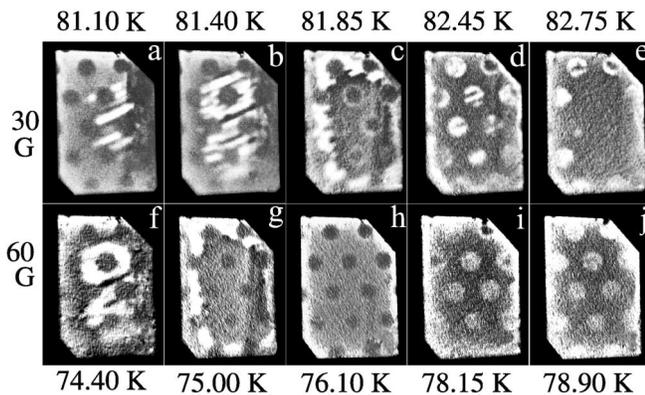


FIG. 2. Differential MO images of the melting process vs T at fields of 30 and 60 G in BSCCO crystal $0.65 \times 0.45 \times 0.01$ mm³. The bright regions are the areas that undergo melting within T modulation of 0.3 K. (a)–(c) and (f)–(h): melting of the pristine regions while the irradiated apertures are still solid. (d), (e), (i), and (j): melting of the irradiated apertures ($B_\phi = 20$ G). The area outside the crystal is blackened [20].

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sample disorder (see [3]). With increasing T [Fig. 2(b)] the liquid expands, remarkably *avoiding* the irradiated apertures. In Fig. 2(c) the entire central pristine part of the sample is liquid, while the apertures with $B_\phi = 20$ G are still solid. In Fig. 2(d) the central apertures melt at 82.45 K, which is about 1 K above T_m of the adjacent pristine regions in Fig. 2(b). The apertures closer to the sample edges begin to melt in Fig. 2(e). The FOT is equally strong in the irradiated and pristine regions: The ΔB step derived from the paramagnetic melting signal [10] is the same in Figs. 2(b) and 2(d). Also the width of the local melting transition is the same within our resolution, i.e., each point in the sample melts within 0.3 K or less. This is the first direct observation of an upward shift of the FOT by correlated disorder.

The melting process at 60 G (Fig. 2, second row) reveals two important differences. First, the shift of the melting temperature, ΔT_m , is about 4 K [difference between Figs. 2(f) and 2(i)] which is much larger as compared to 1 K at 30 G. In Fig. 2(h), for example, the entire pristine sample has melted while the irradiated apertures are still solid. Second, the brightness of the paramagnetic melting signal ΔB in the apertures in Fig. 2(i) is much lower than in the pristine sample, and moreover, the melting in each aperture is broadened over several frames, as exemplified in Fig. 2(j). At still higher fields, above 100 G, no paramagnetic FOT signal is detected in the irradiated apertures with $B_\phi = 20$ G.

We have also used differential MO with field modulation of 1 G (Fig. 3) instead of the T modulation. In addition to detecting the FOT [3], this method provides a very sensitive measurement of the irreversibility line at a very low effective frequency of about 0.1 Hz. Figure 3 demonstrates the determination of the irreversibility line in the region where no FOT is present. In Fig. 3(a) the entire pristine sample is in the liquid state, while the apertures are still solid. The apertures appear black, showing that the external field modulation is shielded due to the enhanced pinning. Upon increasing the field

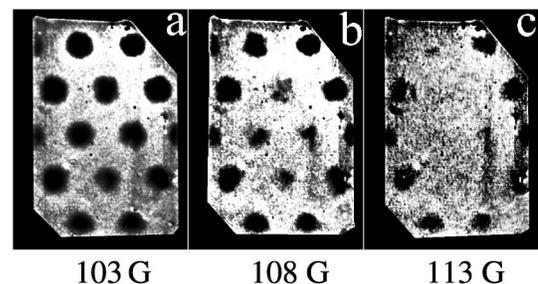


FIG. 3. Differential MO images using field modulation of 1 G in BSCCO crystal of Fig. 2 at three fields at $T = 68.4$ K. (a) All the pristine regions are liquid while the irradiated apertures ($B_\phi = 20$ G) are still solid and irreversible, and hence appear black due to shielding. (b) Partial reversibility of the central apertures. (c) Central apertures are liquid while the apertures closer to the edges begin to melt.

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the black apertures disappear [Figs. 3(b) and 3(c)] revealing the value of the local irreversibility field.

Figure 4 shows the location of the onset of the FOT for $B_\phi = 5, 10, 20,$ and 50 G obtained by T modulation (solid symbols) and of the irreversibility line obtained by field modulation (open symbols). The solid lines are guides to the eye for the FOT lines which terminate at the corresponding critical points (CPs). The irreversibility data coincide with the FOT line below the CP and smoothly extrapolate the location of the transition line to higher fields. The first interesting observation here is that although the structures of the porous vortex matter and of the Bragg glass are very different (Fig. 1), their phase diagrams for $B_\phi = 5$ G in Fig. 4 are almost identical, with a slight upward shift of the FOT. This brings us to an important conclusion that the quasi-long-range order that characterizes the Bragg glass is apparently not an essential requirement for the existence of a FOT [11].

We can understand the upward shift in $B_m(T)$ in Fig. 4 by generalizing the cage model concept. In a pure system each vortex is confined in an elastic potential cage of its neighbors, and melting occurs when the transverse thermal fluctuations $\langle u_T \rangle$ reach a certain fraction c_L of the lattice spacing a_0 . The pores of the matrix are vertical cylinders which provide an additional confining cage potential to the interstitial vortices. This enhanced rigidity reduces $\langle u_T \rangle$, and hence stabilizes the solid crystallites within the pores, thereby shifting $B_m(T)$ upwards. The shift in $B_m(T)$ grows with B_ϕ due to the decrease in the size of the pores. Since the pinning energy of the CDs is usually significantly larger than the elastic energy, the melting of the crystallites may occur without the destruction of the matrix, hence forming an interstitial liquid

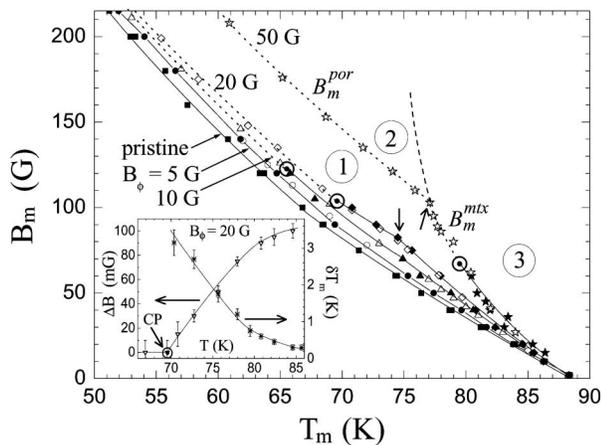


FIG. 4. The melting lines $B_m(T)$ of the pristine and irradiated regions with indicated B_ϕ . Solid (open) symbols are temperature (field) modulation data showing the location of the FOT (irreversibility line). Solid (dotted) lines are guides to the eye of the first-order (continuous) transitions that terminate at the critical points \odot . Inset: The height of the FOT equilibrium magnetization step ΔB which vanishes at the CP, and the local width δT_m of the FOT vs T .

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within a rigid matrix as predicted theoretically [8] and in numerical simulations [12].

Upon increasing the density of CDs the FOT is weakened and it eventually transforms into a continuous transition in Fig. 4. This transformation is seen more clearly by varying the temperature along a given $B_m(T)$ line. The inset to Fig. 4 shows the height of the equilibrium magnetization step ΔB vs T for $B_\phi = 20$ G. As T is decreased ΔB drops and vanishes at a CP below which no discontinuity is found and the melting becomes continuous. In addition, the width of the FOT, δT_m , shows a significant broadening on approaching the CP (Fig. 4 inset, right axis). This indicates that it is a true CP at which the FOT transforms into a continuous transition as expected theoretically [13]. This CP is unlike the apparent point-disorder-induced CP in BSCCO at lower T where no appreciable broadening was observed [14], and where the FOT was found to be obscured by pinning-induced irreversibility in the liquid phase [15]. Here, in contrast, the liquid is fully reversible and no such experimental limitation exists. Figure 4 shows that the CP is shifted to lower fields with B_ϕ [16]. Note that in YBCO the CP was observed to shift upward with CDs [17]. There, however, the CP is induced by the pristine point disorder and the CDs just slightly modify its position by suppressing vortex meandering. In our case, in contrast, the CDs apparently create a new CP that does not exist in pristine crystals.

We now discuss another key finding which is a kink in the $B_m(T)$ lines marked by the arrows in Fig. 4. This kink becomes very prominent when the temperature shift ΔT_m between the irradiated and pristine melting lines is plotted as shown in Fig. 5. The inset displays the corresponding upward shift in field ΔB_m . In BG theory a kink in the transition line is expected to occur at $B_k \approx B_\phi$ [7,8]. We argue that the kink in Fig. 5 is a new feature of a completely different nature, which reflects the collapse of the rigid matrix. The BG kink is experimentally found

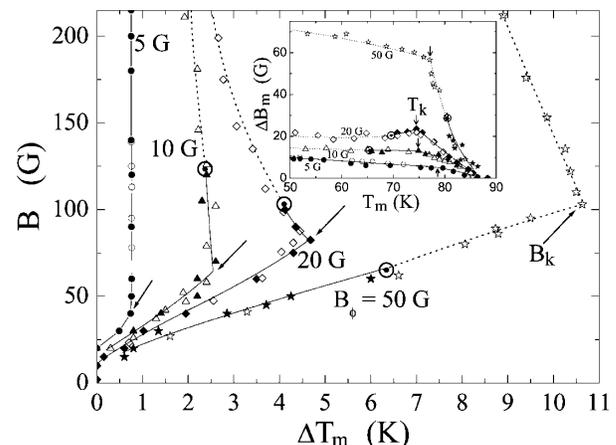


FIG. 5. The shift ΔT_m in the melting temperature for different B_ϕ with respect to the pristine T_m at various fields B . Inset: The upward shift ΔB_m in the melting field vs T_m .

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to occur at B_k/B_ϕ of 1/6 to 1 for large B_ϕ , and B_k usually scales linearly with B_ϕ [18,19]. The kink in Fig. 5 occurs in the opposite regime of $B_k/B_\phi \approx 2$ to 8, and it does not scale with B_ϕ . The BG kink is a broad feature that was found to occur only along a glass transition line [18,19]. In contrast, here the kink is extremely sharp, and it is the first observation of a kink that occurs along a FOT line. Finally, the BG kink is a transition from a strong influence of the CDs below B_k to a much weaker effect of CDs above it [7,8]. Here, in contrast, the situation is just the opposite, as seen in the inset to Fig. 5. The effect of CDs is large at fields above B_k , where the shift ΔB_m is almost constant and approximately equal to B_ϕ , and it collapses rapidly at $T > T_k$. This collapse occurs at $T_k \approx 75$ K, with little dependence on B_ϕ . Interestingly, the kink occurs along the FOT line for $B_\phi \leq 20$ G, while for $B_\phi = 50$ G it falls in the region of continuous transition.

In Fig. 4 three regions are enumerated with respect to $B_\phi = 50$ G curve. In region 1 the crystallites in the pores are stabilized by the rigid matrix and melt at B_m^{por} , well above the pristine $B_m(T)$. In region 2 the pores are liquid while the matrix remains intact, and hence B_m^{por} reflects a melting transition of the softer of the two substances in a heterogeneous medium. This unconventional melting cannot be described by a single set of parameters since it depends on the properties of both the pores and the matrix, each having different field and temperature dependencies. As a result, $B_m^{\text{por}}(T)$ does not extrapolate to T_c , but rather well above it, which is a very uncommon behavior at high temperatures. This heterogeneous process collapses sharply at T_k , above which the matrix delocalizes, resulting in a homogeneous liquid in region 3. The B_m^{mtx} line thus describes the melting process of the matrix due to depinning from the CDs, which leads to immediate melting of the superheated crystallites. Thus in contrast to the common homogeneous BG description, the $B_m^{\text{por}}(T)$ and $B_m^{\text{mtx}}(T)$ lines originate from *two independent heterogeneous processes*, melting of crystallites within a rigid matrix and the collapse of the matrix itself, resulting in a sharp kink at the intersection point. We expect the $B_m^{\text{mtx}}(T)$ line to extend into the liquid phase [8], like the dashed line in Fig. 4, separating the interstitial liquid region 2 from the homogeneous liquid 3. Our MO measurements cannot detect this line since both these regions are fully reversible. In region 2, however, we expect a higher c axis correlation than in region 3, which is apparently consistent with the recently reported recoupling transition [19].

In summary, when vortices outnumber CDs, heterogeneity rather than the average properties of the lattice has to be taken into account for a proper description of the structure and the thermodynamic behavior of the vortex matter. We have found evidence for two mechanisms: melting of superheated crystallites within the pores of a solid matrix and the destruction of the rigid matrix. The intersection point of these two independent processes results in a sharp kink in the observed melting line. The

heterogeneous melting can be either first order or continuous, depending on temperature and the density of CDs. The porous vortex matter may thus provide a tunable model system for general comprehension of melting of nanocrystals in porous materials.

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- [1] P. R. Couchman and W. A. Jesser, *Nature (London)* **269**, 481 (1977); A. N. Goldstein, C. M. Echer, and A. P. Alivisatos, *Science* **256**, 1425 (1992).
- [2] L. Gråbaek *et al.*, *Phys. Rev. Lett.* **64**, 934 (1990); H. W. Sheng *et al.*, *Philos. Mag. Lett.* **73**, 179 (1996); Q. Jiang, Z. Zhang, and J. C. Li, *Chem. Phys. Lett.* **322**, 549 (2000).
- [3] A. Soibel *et al.*, *Nature (London)* **406**, 282 (2000); A. Soibel *et al.*, *Phys. Rev. Lett.* **87**, 167001 (2001); M. Yasugaki *et al.*, *Phys. Rev. B* **65**, 212502 (2002).
- [4] T. Giamarchi and P. Le Doussal, *Phys. Rev. Lett.* **72**, 1530 (1994).
- [5] M. Menghini *et al.*, cond-mat/0210623.
- [6] D. R. Nelson and V. M. Vinokur, *Phys. Rev. Lett.* **68**, 2398 (1992).
- [7] A. I. Larkin and V. M. Vinokur, *Phys. Rev. Lett.* **75**, 4666 (1995).
- [8] L. Radzihovsky, *Phys. Rev. Lett.* **74**, 4923 (1995).
- [9] See EPAPS Document No. E-PRLTAO-90-058307 for color movies of melting at 30 and 50 G. A direct link to this document may be found in the online article's HTML reference section. The document may also be reached via the EPAPS homepage (<http://www.aip.org/pubservs/epaps.html>) or from <ftp.aip.org> in the directory /epaps/. See the EPAPS homepage for more information.
- [10] N. Morozov *et al.*, *Phys. Rev. B* **54**, R3784 (1996).
- [11] S. Colson *et al.*, cond-mat/0204493.
- [12] P. Sen, N. Trivedi, and D. M. Ceperley, *Phys. Rev. Lett.* **86**, 4092 (2001); Y. Nonomura and X. Hu, cond-mat/0212609.
- [13] Y. Imry and M. Wortis, *Phys. Rev. B* **19**, 3580 (1979).
- [14] E. Zeldov *et al.*, *Nature (London)* **375**, 373 (1995).
- [15] N. Avraham *et al.*, *Nature (London)* **411**, 451 (2001).
- [16] B. Khaykovich *et al.*, *Phys. Rev. B* **57**, R14088 (1998).
- [17] W. K. Kwok *et al.*, *Phys. Rev. Lett.* **84**, 3706 (2000).
- [18] C. J. van der Beek *et al.*, *Phys. Rev. Lett.* **86**, 5136 (2001); L. Krusin-Elbaum *et al.*, *Phys. Rev. Lett.* **72**, 1914 (1994).
- [19] M. Sato *et al.*, *Phys. Rev. Lett.* **79**, 3759 (1997); T. Tamegai *et al.*, in *Advances in Superconductivity XI: Proceedings of the 11th International Symposium on Superconductivity (ISS'98), 1998, Fukuoka* (Springer-Verlag, Tokyo, 1999), p. 571.
- [20] Color MO movies are available at <http://www.weizmann.ac.il/home/fnsup/>